Table 2(b) shows, the uranium-atom position corresponded remarkably well to the correct one. Reeke & Lipscomb (1969) have also reported that the tangent formula works at 6.0 Å.

The 8.0 Å data, however, did not give as good a result. Various attempts at multi-solution phasing with ΔE greater than 1.0 and 0.5 produced a number of phase sets, only two of which had R less than 0.40. The less consistent of the two had the lower R of 0.38, by nearly 0.02, and was the better solution, as judged by the ΔE map. Fig. 1(b) [Table 2(b)] shows that the uranium atom, determined at 8.0 Å resolution was over 4 Å away from its true position. The ΔE map did not show the good peak discrimination encountered earlier; the strongest spurious peak had half the strength of the correct one. It is possible that significant errors in the ΔE values at this low resolution were to some extent responsible for these results.

It thus seems that direct three-dimensional phasing of ΔF data sets can reliably locate heavy atoms in biological macromolecules, even when only 5 or 6 Å resolution data are available, provided that the individual heavy atoms are well resolved. Not too much reliance can be placed on lower-resolution results. As Steitz (1968) has pointed out, these methods could be of considerable use in the structure analysis of molecules with multiple heavy-atom binding sites. Where high-symmetry space groups are involved, Patterson map interpretation becomes especially difficult

and statistical phasing to locate heavy atoms is particularly advantageous.

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La₂O₂S structure refinement and crystal field.* By B. MOROSIN, Sandia Laboratories, Albuquerque, New Mexico 87115, U.S.A. and D. J. NEWMAN, Department of Physics, Queen Mary College, Mile End Road, London E1 4NS, England

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Parameters on the structure of La_2O_2S have been refined by the least-squares method using 396 Mo K α intensity data. In space group $P\overline{3}m1$, the La^{3+} ion occupies 3(m) sites surrounded below by three S^{2-} ions (3.037 Å) and above by three O^{2-} ions (2.424 Å) and one axial O^{2-} ion (2.423 Å). A new crystal-field analysis has been performed using this data.

Introduction

The rare-earth oxysulfides activated with trivalent rareearth ions form an important technological class of materials with high luminous efficiency (Struck & Fonger, 1971; Dobrov & Buchanan, 1972). In particular, neodymiumactivated lanthanum oxysulfide, La2O2S:Nd, has received some attention as a promising new high-gain laser material and, hence, has stimulated studies on the growth of the required single crystals (Alves, Buchanan, Wickersheim & Yates, 1971; Baughman, 1973). The simplicity of the crystal structure as well as the many various isomorphous members and/or dopants makes the material ideal for analysis of crystal-field parameters (Sovers & Yoshioka, 1969; Newman & Stedman, 1971). Unfortunately, only approximate atomic parameters obtained by the powder method were available (Wyckoff, 1960) leading Newman & Stedman to obtain values for superposition model parameters (in their crystal-field analysis) which were difficult to interpret. Crystal structure parameters determined on 1 wt. % Nd doped

 La_2O_2S material kindly supplied by Baughman are reported in this note.

Experimental details and results

A single-crystal specimen was ground to a radius of 0.0121 cm. The specimen was examined by long-exposure photographic methods to insure that $P\overline{3}m1$ is the correct space group. Lattice constants were determined to be $a_0 = 4.049$ (1) and $c_0 = 6.939$ (2) Å by measurements made on a Picker diffractometer. The θ -2 θ scan technique and a scintillation detector employing pulse-height discrimination were used to measure a complete hemisphere (to 95° 2 θ) of Mo K α intensity data. A unique data set of 396 intensities was obtained by averaging the symmetry-related values provided the differences were less than σ_{ave}/\sqrt{n} , where σ_{ave} is the average σ for *n* measurements with the usual definition of $\sigma = (N_{\rm SC} + K^2 N_B)^{1/2}$, where $N_{\rm SC}$, N_B and K are the total scan count, background counts and the time ratio of the scan to background respectively. In the few instances where an intensity did not agree to within this criterion, it was discarded and averaging was performed on the remaining intensities. All reflections were considered observed. Spherical absorption corrections (μ [Mo K α] = 190 cm⁻¹) were ap-

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plied from Table 5.3.6B of International Tables for X-ray Crystallography (1969). Initial parameters were taken from Wyckoff and the intensity data subjected to least-squares refinement. The function, $w(F_o - F_c)^2$, was minimized with $w = n/\sigma_{ave}^2$ and with structure factors calculated using scattering factors taken from Table 3.3.1A and 3.3.1B and dispersion corrections from Table 3.3.2C of International Tables for X-ray Crystallography (1962). The residual $R = \sum ||F_o| - F_c|| / \sum F_o|$ was 0.038 and 0.031, respectively, using isotropic and anisotropic thermal parameters. The positional and thermal parameters, observed and calculated structure factors, and interatomic separations are given in Tables 1, 2 and 3 respectively.

In this structure, the rare-earth ion is located on a threefold axis with a triangle of three sulfide ions below it (3.037 Å) and a triangle of three oxide ions (2.424 Å) and one axial oxide ion (2.423 Å) above it. The sulfide ion is surrounded by two triangles (one above and one below) of three oxide ions at 2.481 Å and two triangles of rareearth ions while the oxide ion is surrounded below by one axially located rare-earth ion and a triangle of three oxide ions (2.940 Å) and above by a triangle each of sulfide and rare-earth ions. These separations are slightly different from the corresponding values (3.086, 2.430 and 2.389 Å) respec-

Table 1. Positional and thermal parameters for La₂O₂S:Nd

	х	У	z	u_{11}	l1 ₃₃
S	0	0	0	0.29 (6)	0.37 (8)
La	$\frac{2}{3}$	13	0.27929 (6)	0.15(2)	0.31(2)
0	$\frac{2}{3}$	$\frac{1}{3}$	0.6287 (8)	0.17 (9)	0.19 (14)

Table 2. Observed and calculated structure factorsfor La2O2S

C+7+L	9.3.6	4 270 265	1 144 134	2 36 53	6 3/9 385	1 194 188	1 276 275	11 241 101	
		5 233 225	4 433 429	8 344 370	5 310 312		* 310 301		0 100 103
1 134 141	C 593 598	6 114 117	5 52 52	9 210 214	6 152 158	3.0.1	5 252 250		100 100
2 449 478	1 18 84	7 1.2 341	6 551 527	• 210 210	7 416 411		6 128 128	*****	0 203 210
3 696 66	2 455 453		7 84 68		# 120 111		7 119 115		7 1 22 141
6 680 461	3 191 183		8 165 160	1.0.1	9 141 340		8 85 84	1 548 544	
5 606 186	6 612 mus	0.2.1	9 249 297		1 125 121	2 420 433	9 180 148	2 341 331	3.2.L
3 400 300		0.7.0		0 151 155	11 12 121	3 346 383	4 504 589	3 234 237	
0 1 21 703			10 210 214	1 343 333	11 727 328	- 381 -07		4 114 313	1 62 51
1 322 490	0 101 100	0 134 140		\$ 110 106	14 101 112	> 300 324	34441	5 34 39	2 240 245
8 107 144		1 :48 140	12 02 09	3 323 317		0 171 100		6 187 195	1 226 233
4 461 401	9 114 110	5 520 520	11 200 275	- 128 108	2.3.6	1 412 413	5 108 tor	/ 01 74	+ 255 255
10 116 144	4 554 154	1 1/6 170		> 318 316		8 119 118	1 231 221	8 273 276	> 2.1 218
11 388 382	10 110 177		1	6 156 162	0 510 514	4 342 354	2 298 291		0 108 111
12 190 195	11 340 340	1.0.1			1 286 285	10 123 127	1 209 208		1 274 241
13 169 11	12 174 177		1 230 241		2 188 184	11 124 14	4 292 28:		
14 258 265	13 136 136	1 678 551	; 310 -13	2.0.1	\$ 219 217		3 8 38	4.3.6	3.3.4
		1 205 204	2 185 183		+ 102 156	16 10 11	A 154 154		
3 . 1 . 1	24444	1 682 663	1 + 70 +00	1 113 364	> >1 41	19 174 170	7 58 51	1 304 301	1 1 1 1 1 1 1 1
			A 141 19-4	2 441 481	N 649 644			2 118 115	1 142 144
1 160 110	1 1 9 9 2 2 2	4 201 270	5 6 7 8 - 6 4	3 104 180	2 41 44	1.1.4		3 346 338	2 258 250
	1 240 121	3 346 134		4 440 450			3.3.6	4 122 117	3 171 178
1 343 107		6 311 548	0 220 272	5 59 54		1 302 417		5 138 330	
2 711 484		1 542 542	1 (10 (10	0 774 771		2 414 472	15/1 146	6 170 172	6+2+6
3 425 406	, , , , , , , , , , , , , , , , , , , ,	8 430 415		1 11 12	10 212 712	3 300 311	1 322 313	7 126 122	
4 501 408	• 377 37:	9 227 224	9 174 182	4 376 377	11 217 218	+ 370 18¥	2 102 97		1 59 53
5 75 70	> 42 42	10 320 316	10 211 210	9 11. 112		5 71 65	3 10: 299	A	2 253 305
6 675 574	0 473 403	11 23 13	11 30 15		2.4.6				1 211 2.5
7 79 77	7 67 61	12 365 158	12 110 111	11 253 256		1 00 01	4.0.0		* 2×1 2×2
8 405 346	8 1:1 319	13 60 50		11 77 776	v 192 185			0 3/3 380	5 3 3 4 3 3 4
9 115 128	9 257 260	14 314 343	1.0.0	12 81 82	1 421 424	6 31 7 772	1.085.087	1 61 43	1 11 11 11 11
13 110 101	10 217 241			11 286 285	2 138 133	4 201 211		2 274 268	0 110 117
11 210 244	11 222 275		326 506		1 107 144	12 544 521		3 234 238	1 542 1.1
		1.1.1	1	2.1.1		11 224 211		4 234 224	8 67 77
13 304 285	1.2.		2 174 175		5 142 124	12 78 16	· 100 10v		
13 304 243		2 995 954	1 1 1 1 1	1 536 552	, 30/ 3.0		7 462 448	5121.	6111
14 30 32		1 103 119	1 311 104	2 205 223	V 184 184	3+2+1	6 217 221		
	3 185 195	2 472 493	6 345 331	3 512 513	142 144		1 221 220	1 255 260	1 200 200
0.2.1	1 438 426	3 439 437	5 215 211	\$ 235 210	5 275 280	1 473 444	8 126 128	1 111 100	2 244 269
	2 143 141	4 403 440	6 1 18 138	5 525 516	9 144 147	2 137 137	9 159 174	1 105 100	1 10% 192
0 274 313	3 401 301	5 379 306	7 374 363	A 255 259		3 434 423	10 219 261	6 121 124	4 271 204
1 554 567	4 148 142	6 187 184	4 47 90	1 1 1 1 1 1 1	2.7.1	4 153 159	1 1 12	• 3/1 3/6	5 14 12
2 252 254	5 672 394	7 691 676	9 114 107	· · · · · · · · ·		6 6 3 3 6 3 9		2 39 40	
3 563 536	6 193 196	. 130 130	15 1.4 134	1 341 310	J 615 199			6 422 410	6 364 116
	2 232 231	0 10 10	11 291 996	4 19/ 501	1 65 51	6 205 2:2		' 61 55	
	4 101 291	4 340 384		1 2 2 9 2 2 9 1	2 103 344	1 212 217		8 212 287	6.2.
	¥ 15. 153	17 131 141	1.5.0	10 10 14	1 246 211	8 311 315		4 210 232	
3 210 211		11 372 300		12 114 313		9 102 100	2 301 312	10 205 216	1 302 301
1 200 214	1- 510 510	15 199 188		13 54 48	* 20* 277	10 276 273	1 102 104		
8 412 395		13 :06 100	2 176 178		> 221 218	11 46 15	~ 31 - 131	24141	1.0.4
9 513 515	C.0.L		1 243 238	2.2.1	0 112 111		5 255 271		
10 305 303		1.2.4	2 328 316		\$ 500 541	1.3.4	0 129 130	1	1 304 113
11 33 13	0 615 616		3 210 221				1 304 303	1 1 1 1 1 1 1	2 83 97
12 350 345	1 58 51	0 263 284	 109 301 	1 1 1 1	2.0.1		8 88 96	1 141 171	1 271 294
13 50 44	2 313 405	1 141 151	5 43 38			1 42 52	9 242 40*		
	3 243 242	2 662 666	6 106 381	2 428 410	0 140 147	1	1 107 106	P 11A 154	
				> >>> >>>		e ,,: ,.e.	100	N 147 104	

Table 3. Interatomic separations and angles for La₂O₂S:Nd

La-O'	2·423 (6) Å	SLa-S	83.62 (1)
La-O''	2.424 (2)	O'La-S	129.66 (1)
La-S	3.037 (1)	O'LaO''	74.7 (1)
La-La	3.853 (1)	O''La-O''	113.29 (9)
SO'	3.481 (1)	O'SLa	87.47 (7)
0'-0''	2.940 (6)	LaSLa	83.62 (1)
		O'SO'	71.14 (8)
		LaSO'	43.00 (2)

tively) calculated from the parameters in Wyckoff. The associated changes in angular coordination are, however, in the correct direction to yield improved accuracy for the intrinsic parameters of the sulfide ion determined by the superposition model in the analysis of the crystal field in rare-earth oxysulfides by Newman & Stedman.

To demonstrate this the angles given in Table 3 are used to recalculate the intrinsic parameters for the system La_2O_2S : Eu. The method of analysis has been described by Newman & Stedman and the experimental crystal-field parameters were obtained by Sovers & Yoshioka. The following crystal field parameters representing the contributions of single oxygen or sulphur ligands are obtained:

$$\bar{A}_4(O) = 88.2 \text{ cm}^{-1}$$

 $\bar{A}_4(S) = 18.8 \text{ cm}^{-1}$
 $\bar{A}_5(S) = 5.1 \text{ cm}^{-1}.$

The main difference between these parameters and those determined by Newman & Stedman is the much reduced value of $\bar{A}_4(S)$ (originally given as 46.9 cm⁻¹ for the Gd₂O₂S host). The ratio $\bar{A}_4(S)/\bar{A}_6(S) = 3.7$ is now very similar to that found in other systems. Low values of the sulphur parameters are consistent with the large La–S separations in this crystal. These results show that accurate structural information is an essential prerequisite for obtaining reliable intrinsic parameters using the superposition model.

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